



INTERNATIONAL APPLICATION PUBLISHED UNDER THE PATENT COOPERATION TREATY (PCT)

(51) International Patent Classification ⁵ : C22C 16/00	A1	(11) International Publication Number: WO 93/16205 (43) International Publication Date: 19 August 1993 (19.08.93)
(21) International Application Number: PCT/US92/06142 (22) International Filing Date: 24 July 1992 (24.07.92) (30) Priority data: 835,348 14 February 1992 (14.02.92) US (71) Applicant: COMBUSTION ENGINEERING, INC. [US/ US]; 1000 Prospect Hill Road, Windsor, CT 06095 (US). (72) Inventor: GARDE, Anand, Madhav ; 2 Sherwood Lane, West Simsbury, CT 06092 (US). (74) Agents: MULHOLLAND, John, H. et al.; ABB Business Services Inc., Intellectual Property Law Dept., 1000 Pro- spect Hill Road, Windsor, CT 06095 (US).		(81) Designated States: KR, UA, European patent (AT, BE, CH, DE, DK, ES, FR, GB, GR, IT, LU, MC, NL, SE). Published <i>With international search report.</i>
(54) Title: CREEP RESISTANT ZIRCONIUM ALLOY (57) Abstract A zirconium alloy which imparts good creep strength, while also providing favorable neutron cross section, improved corrosion resistance, low hydrogen uptake and good fabricability is described which contains vanadium in a range of from a measurable amount up to 1.0 wt%, wherein either limit is typical; niobium in a range of from a measurable amount up to 1.0 wt%, wherein either limit is typical; antimony in a range of from a measurable amount up to 0.2 wt%, wherein either limit is typical; tellurium in a range of from a measurable amount up to 0.2 wt%, wherein either limit is typical; tin in a range of from a measurable amount up to 0.5 wt%, wherein either limit is typical; iron in a range of 0.2 to 0.5 wt%, typically 0.35 wt%; chromium in a range of from 0.1 to 0.4 wt%, typically 0.25 wt%; silicon in a range of 50 to 200 ppm, wherein either limit is typical; and oxygen in a range of from a measurable amount up to 2200 ppm, wherein either limit is typical and the balance zirconium.		

FOR THE PURPOSES OF INFORMATION ONLY

Codes used to identify States party to the PCT on the front pages of pamphlets publishing international applications under the PCT

AT	Austria	FR	France	MR	Mauritania
AU	Australia	GA	Gabon	MW	Malawi
BB	Barbados	GB	United Kingdom	NL	Netherlands
BE	Belgium	GN	Ghana	NO	Norway
BF	Burkina Faso	GR	Greece	NZ	New Zealand
BG	Bulgaria	HU	Hungary	PL	Poland
BJ	Benin	IE	Ireland	PT	Portugal
BR	Brazil	IT	Italy	RO	Romania
CA	Canada	JP	Japan	RU	Russian Federation
CF	Central African Republic	KP	Democratic People's Republic of Korea	SD	Sudan
CG	Congo	KR	Republic of Korea	SE	Sweden
CH	Switzerland	KZ	Kazakhstan	SK	Slovak Republic
CI	Côte d'Ivoire	LI	Liechtenstein	SN	Senegal
CM	Cameroon	LK	Sri Lanka	SU	Soviet Union
CS	Czechoslovakia	LU	Luxembourg	TD	Chad
CZ	Czech Republic	MC	Monaco	TG	Togo
DE	Germany	MG	Madagascar	UA	Ukraine
DK	Denmark	ML	Mali	US	United States of America
ES	Spain	MN	Mongolia	VN	Viet Nam
FI	Finland				

CREEP RESISTANT ZIRCONIUM ALLOY

BACKGROUND OF THE INVENTION

This invention relates to alloys for use in light water nuclear reactor (LWR) core structural components and fuel claddings. More particularly, this invention relates to a zirconium alloy with second phase vanadium precipitates which are stable with respect to neutron exposure and high temperature exposure. Still more particularly, this invention relates to a zirconium alloy having stable second phase vanadium precipitates, while containing tin levels below that of conventional zirconium alloys and various additional alloying elements. This alloy is designed to function at high coolant temperatures and discharge burn-ups and to provide acceptable levels of creep resistance, neutron cross section, corrosion resistance, hydrogen uptake and fabricability.

DESCRIPTION OF THE PRIOR ART

Zirconium alloys are used in fuel rod claddings and in fuel assembly structural components of nuclear reactors (e.g., guide or thimble tubes, grid strips, instrument tubes, and so forth) because they exhibit a low neutron cross section, good corrosion resistance against high pressure/high temperature steam and water, and good mechanical strength and fabricability. Zirconium alloys, particularly those commonly known as Zircaloy-2 and Zircaloy-4, have also been used in LWR cores because of their relatively small capture cross section for thermal neutrons. "Zircaloy" is a common name for zirconium-tin alloys. Zircaloy-

- 2 -

4, for example, has 0.18 to 0.24 percent by weight (wt%) iron, 0.07 to 0.13 wt% chromium, oxygen in the range of from 1000 to 1600 ppm, 1.2 to 1.7 wt% tin, and the remainder zirconium.

The addition of 0.5 to 2.0 wt% niobium, up to 1.5 wt% tin and up to 0.25 wt% of a third alloying element to zirconium alloys for purposes of corrosion resistance in the reactor core is suggested in U.S. Patent No. 4,649,023 as part of a teaching for producing a microstructure of homogeneously disbursed fine precipitates of less than about 800 Å. The third alloying element is a constituent such as iron, chromium, molybdenum, vanadium, copper, nickel and tungsten.

U.S. Patent No. 5,023,048 describes a fuel rod comprising a cladding tube having an inner tubular layer and an outer surface layer composed of differing zirconium alloys. The inner tubular layer is made from a conventional zirconium alloy such as Zircaloy-4. The outer surface layer is made from a zirconium alloy containing 0.35 to 0.65 wt% tin, 0.2 to 0.65 wt% iron, 0.09 to 0.16 wt% oxygen, and 0.35 to 0.65 wt% niobium or 0.25 to 0.35 wt% vanadium.

Recent trends in the nuclear industry include shifts toward higher coolant temperatures to increase thermal efficiency and toward higher fuel discharge burn-ups to increase fuel utilization. Both the higher coolant temperatures and the higher discharge burn-ups tend to dissolve second phase particles in conventional Zircaloys, and thereby decreasing the creep resistance of these materials. Moreover such conditions increase in-reactor corrosion and hydrogen uptake. Unfortunately, when

- 3 -

the level of tin is lowered to improve corrosion resistance for these applications, the creep resistance of these materials is further degraded due to the loss of solid solution hardening.

Accordingly, it is a continuing problem in this art to develop a zirconium alloy having superior creep strength, while providing good corrosion resistance as well as low neutron absorption, reduced hydrogen absorption by the alloy and good fabricability.

SUMMARY OF THE INVENTION

It is, therefore, an object of this invention to provide a zirconium alloy with vanadium precipitates which are stable with respect to neutron exposure as well as high temperature exposure.

It is another object of this invention to provide a zirconium alloy having tin levels below that of conventional Zircalloys.

It is an additional object of this invention to provide a zirconium alloy with an improved creep resistance while maintaining reasonable levels of low neutron cross section, corrosion resistance, low hydrogen uptake and good fabricability.

It is an additional object of this invention to provide a zirconium alloy comprising vanadium (V) in a range of from a measurable amount up to 1.0 wt%, wherein either limit is typical; niobium (Nb) in a range of from a measurable amount up to 1.0 wt%, wherein either limit is typical; antimony (Sb) in a range of from a measurable amount up to 0.2 wt%, wherein either limit is

- 4 -

typical; tellurium (Te) in a range of from a measurable amount up to 0.2 wt%, wherein either limit is typical; tin (Sn) in a range of from a measurable amount up to 0.5 wt%, wherein either limit is typical; iron (Fe) in a range of 0.2 to 0.5 wt%, typically 0.35 wt%; chromium (Cr) in a range of from 0.1 to 0.4 wt%, typically 0.25 wt%; silicon (Si) in a range of 50 to 200 parts per million (ppm), wherein either limit is typical; oxygen (O) in a range of from a measurable amount up to 2200 ppm, wherein either limit is typical; and the balance zirconium (Zr).

DESCRIPTION OF THE PREFERRED EMBODIMENT

The invention is based upon the theory that, because of its limited solubility, vanadium will precipitate as ZrV_2 and that such precipitates will impart good creep resistance, resist coarsening, exhibit low hydrogen uptake, and be stable under neutron flux and at high burnups. Moreover, based on available creep data⁽¹⁾, it is theorized that a complex alloy containing many alloying elements, both in solid solution as well as in stable second phase particles, should have superior creep resistance when compared to simple alloys. The reasons for selecting specific levels of various alloying elements are given below, and the composition of the alloy according to an embodiment of the present invention is shown in Table 1.

The zirconium alloy of the present invention, therefore, includes vanadium (V) in a range of from a measurable amount up to 1.0 wt%, wherein either limit is typical; niobium

- 5 -

(Nb) in a range of from a measurable amount up to 1.0 wt% wherein either limit is typical; antimony (Sb) in a range of from a measurable amount up to .2 wt%, wherein either limit is typical; tellurium (Te) in a range of from a measurable amount up to 0.2 wt%, wherein either limit is typical; tin (Sn) in a range of from a measurable amount up to 0.5 wt%, wherein either limit is typical; iron (Fe) in a range of 0.2 to 0.5 wt%, typically 0.35 wt%; chromium (Cr) in a range of from 0.1 to 0.4 wt%, typically 0.25 wt%; silicon (Si) in a range of 50 to 200 ppm wherein either limit is typical; oxygen (O) in a range of from a measurable amount up to 2200 ppm, wherein either limit is typical; and the balance zirconium (Zr).

Vanadium, in a range of from a measurable amount to 1.0 wt%, is added as an alloying element to reduce hydrogen uptake.⁽²⁾ Moreover, due to the fact that the densities of zirconium and vanadium are very close to one another, precipitation of ZrV_2 should result in second phase particles that are coherent and will not coarsen or dissolve easily. Finally, additions of vanadium up to 0.4 wt% in zirconium-iron binary alloys has been shown to result in corrosion resistance superior to Zircaloy-4.⁽³⁾

Niobium, in an amount from a measurable amount to 1.0 wt%, is added to improve the corrosion resistance,⁽⁴⁾ to improve the irradiated ductility,⁽⁵⁾ to reduce the hydrogen absorption,⁽⁵⁾ and to increase creep resistance of the new alloy.⁽⁶⁾ In concentrations beyond 0.5 wt%, beta niobium will precipitate, with neutron irradiation possibly causing additional

- 6 -

precipitation.⁽⁷⁾ Niobium also stabilizes irradiated dislocation structures with the formation of niobium-oxygen radiation defect complexes.

Antimony and tellurium, added in amounts ranging from a measurable amount up to 0.2 wt%, decrease the hydrogen uptake by the alloy.⁽⁸⁾ Since the densities of both antimony and tellurium are very close to that of zirconium, second phase particles, if they precipitate, will not coarsen easily.

A decrease in the tin level below the 1.2 wt% lower limit found in Zircaloy-4 improves its corrosion resistance.⁽⁹⁾ However, the trend of the mechanical property data regarding the influence of tin content on the thermal creep of zirconium alloys at 400°C indicates that a decrease in tin level will degrade the creep resistance of zirconium alloys.⁽¹⁰⁾ The selected range of tin level of from a measurable amount up to 0.5 wt% requires that additional alloying elements be added to prevent such degradation.

The corrosion resistance of Zircaloy-2 and iron alloys in both 360°C water and 400°C steam depends on the iron level.⁽¹¹⁾ While the best corrosion resistance in 360°C water was observed with 0.45 wt% iron, the best corrosion resistance in 400°C steam was observed at 0.25 wt% iron. Therefore, iron is added in a range of from 0.2 to 0.5 wt%. In order to achieve good corrosion resistance in both steam and water environments, a preferable intermediate value of 0.35 percent iron may be selected for the new alloy of the invention.

- 7 -

Chromium, in the range of 0.1 to 0.4 wt% and typically 0.25 wt%, is added to optimize the corrosion resistance of the new alloy.

Silicon, in a range of 50 to 200 ppm is added as an alloying element to reduce the hydrogen absorption by the alloy and to reduce variations in the corrosion resistance with variations in the processing history of the alloy.⁽⁹⁾

Oxygen, in a range of from a measurable amount up to 2220 ppm, is added as a solid solution hardening alloying element.

As previously stated, zirconium is desirable as a bulk material due to its favorable neutron cross section, corrosion resistance, mechanical strength and fabricability.

Thus, by its selected composition, the invention of the new alloy described in this disclosure achieves stable second phase particles, which impart good creep resistance, while maintaining low neutron cross section, good corrosion resistance, reduced hydrogen absorption and good fabricability. The exposure of known zirconium alloys to a water reactor environment results in irradiation damage to the second phase particles. This reduces the creep resistance of the irradiated alloys. Moreover, by lowering the tin level to improve corrosion resistance, creep resistance is likewise reduced. A new zirconium alloy, according to this invention, with optimum levels of vanadium, niobium, antimony, tellurium, iron, chromium, silicon, oxygen and tin is proposed to overcome these problems.

- 8 -

BIBLIOGRAPHY

(1) Grigoriev, V.M., Nikulina, A.V. and Peregud, M.M., "Evolution of Zr-Nb Base Alloys for LWR Fuel Clads," paper presented at the IAEA Technical Committee Meeting on Fundamental Aspects of Corrosion of Zirconium-Base Alloys for Water Reactor Environments, Portland, Oregon, September 11-15, 1989.

(2) Parfenov, B.G., Gerasimov, V.V. and Venediktova, G.I., Corrosion of Zirconium and Zirconium Alloys, Israel Program for Scientific Translations, Jerusalem, p. 119 (1969).

(3) Charquet, D., Gros, J.P., and Wadier, J.F., "The Development of Corrosion Resistant Zirconium Alloys," Proceedings of the International ANS-ENS Topical Meeting on LWR Fuel Performance, Avignon, France, April 21-24, 1991, Vol. 1, pp. 143-152.

(4) Isobe, T. and Matsuo, Y., "Development of High Corrosion Resistance Zirconium-base Alloys," Zirconium in the Nuclear Industry, 9th International Symposium, ASTM STP 1132, C.M. Eucken and A. M. Garde, Eds., American Society for Testing and Materials, Philadelphia, 1991, pp. 346-367.

(5) Garde, A.M., U.S. Patent No. 4,879,093, "Ductile Irradiated Zirconium Alloy," issue date November 7, 1989.

(6) Fuchs, H.P., Garzarolli, F., Weidinger, H.G., Bodmer, R.P., Meier G., Besch, O.-A. and Lisdat, R., "Cladding and Structural Material Development for the Advanced Siemens PWR Fuel 'FOCUS'," Proceedings of the International ANS-ENS Topical Meeting on LWR Fuel Performance, Avignon, France, April 21-24, 1991, Vol. 2, pp. 682-690.

- 9 -

(7) Urbanic, V.F. and Gilbert, R.W., "Effect of Microstructure on the Corrosion of Zr-2.5Nb Alloy," paper presented at the IAEA Technical Committee Meeting on Fundamental Aspects of Corrosion of Zirconium-Base Alloys for Water Reactor Environments, Portland, Oregon, September 11-15, 1989.

(8) Garde, A.M., U.S. Patent No. 5,080,861, "Zirconium Alloy with Superior Corrosion Resistance at Extended Burnups," issue date January 14, 1992.

(9) Eucken, C.M., Finden, P.T., Trapp-Pritsching, S. and Weidinger, H.G., "Influence of Chemical Composition on Uniform Corrosion of Zirconium Base Alloys in Autoclave Tests," Zirconium in the Nuclear Industry Eighth International Symposium, ASTM STP 1023, L.F.P. Van Swam and C.M. Eucken, Eds; American Society for Testing and Materials, Philadelphia, 1989, pp. 113-127.

(10) McInteer, W.A., Baty, D.L. and Stein, K.O., "The Influence of tin content on the Thermal creep of Zircaloy-4," Zirconium in the Nuclear Industry, Eighth International Symposium, ASTM STP 1023, L.F.P. Van Swam and C.M. Eucken, Eds; American Society for Testing and Materials, Philadelphia, 1989, pp. 621-640.

(11) Scott, D.B., "Notes on the Corrosion Behavior of Zircaloy-2 with Various Levels of Iron Content," Zirconium Highlights, WAPD-ZH-24, p. 11 (1960).

- 10 -

TABLE 1Preferred Embodiment of the Zirconium Alloy

	<u>Range</u>	<u>Typical</u>
Vanadium, wt%	Measurable amount up to 1.0%	same
Niobium, wt%	Measurable amount up to 1.0%	same
Antimony, wt%	Measurable amount up to 0.2%	same
Tellurium, wt%	Measurable amount up to 0.2%	same
Tin, wt%	Measurable amount up to 0.5%	same
Iron, wt%	0.2 to 0.5%	0.35%
Chromium, wt%	0.1 to 0.4%	0.25%
Silicon, ppm	50 - 200 ppm	same
Oxygen, ppm	Measurable amount up to 2200 ppm	same

- 11 -

IN THE CLAIMS

1. A zirconium alloy for use in light water nuclear core structure elements and in fuel cladding, which comprises a composition as follows:

vanadium, in a range from a measurable amount up to 1.0 wt%;

niobium, in a range from a measurable amount up to 1.0 wt%;

antimony, in a range from a measurable amount up to 0.2 wt%

tellurium, in a range from a measurable amount up to 0.2 wt%;

tin, in a range of from a measurable amount up to 0.5 wt%;

iron, in a range of 0.2 to 0.5 wt%;

chromium, in a range of 0.1 to 0.4 wt%;

silicon, in a range of 50 to 200 ppm;

oxygen, in a range of from a measurable amount up to 2200 ppm; and

zirconium, constituting the balance of said composition.

2. The alloy as set forth in claim 1, wherein said chromium concentration is about 0.25 wt%.

3. The alloy as set forth in claim 1, wherein said iron concentration is about 0.35 wt%.

- 12 -

4. A method of making a zirconium alloy comprising the steps of:

providing a zirconium alloy having niobium, in a range from a measurable amount up to 1.0 wt%; antimony, in a range from a measurable amount up to 0.2 wt%; tellurium, in a range from a measurable amount up to 0.2 wt%; tin, in a range of from a measurable amount up to 0.5 wt%; iron, in a range of 0.2 to 0.5 wt%; chromium, in a range of 0.1 to 0.4 wt%; silicon, in a range of 50 to 200 ppm; oxygen, in a range of from a measurable amount up to 2200 ppm; and zirconium, constituting the balance of said composition; and

adding vanadium, in a range from a measurable amount up to 1.0 wt% as an alloying agent to reduce hydrogen uptake, increase corrosion resistance and provide stable second phase particles.

5. The method as set forth in claim 4, wherein said chromium concentration is about 0.25 wt%.

6. The method as set forth in claim 4, wherein said iron concentration is about 0.35 wt%.

I. CLASSIFICATION OF SUBJECT MATTER (if several classification symbols apply, indicate all) ⁶		
According to International Patent Classification (IPC) or to both National Classification and IPC		
Int.Cl. 5 C22C16/00		
II. FIELDS SEARCHED		
Minimum Documentation Searched ⁷		
Classification System	Classification Symbols	
Int.Cl. 5	C22C ; G21C	
Documentation Searched other than Minimum Documentation to the Extent that such Documents are Included in the Fields Searched ⁸		
III. DOCUMENTS CONSIDERED TO BE RELEVANT⁹		
Category ¹⁰	Citation of Document, ¹¹ with indication, where appropriate, of the relevant passages ¹²	Relevant to Claim No. ¹³
A	FR,A,2 626 291 (MITSUBISHI KINZOKU K.K.) 28 July 1989 *Claims 1-3 ; page 8, Table 1-1 *	1-6
A	DE,A,3 805 124 (SIEMENS AG) 31 August 1989 see the whole document	1-6
A	EP,A,0 468 093 (COMBUSTION ENGINEERING) 29 January 1992 *Claims 1-4*	1-6
A	& US,A,5 080 861 cited in the application	
A	US,A,2 872 310 (WILHELM ET AL.) 3 February 1959 see the whole document	1-6
<p>¹⁰ Special categories of cited documents : ¹⁰</p> <p>"A" document defining the general state of the art which is not considered to be of particular relevance</p> <p>"E" earlier document but published on or after the international filing date</p> <p>"L" document which may throw doubts on priority claim(s) or which is cited to establish the publication date of another citation or other special reason (as specified)</p> <p>"O" document referring to an oral disclosure, use, exhibition or other means</p> <p>"P" document published prior to the international filing date but later than the priority date claimed</p> <p>"T" later document published after the international filing date or priority date and not in conflict with the application but cited to understand the principle or theory underlying the invention</p> <p>"X" document of particular relevance; the claimed invention cannot be considered novel or cannot be considered to involve an inventive step</p> <p>"Y" document of particular relevance; the claimed invention cannot be considered to involve an inventive step when the document is combined with one or more other such documents, such combination being obvious to a person skilled in the art.</p> <p>"&" document member of the same patent family</p>		
IV. CERTIFICATION		
Date of the Actual Completion of the International Search	Date of Mailing of this International Search Report	
13 OCTOBER 1992	29. 10. 92	
International Searching Authority	Signature of Authorized Officer	
EUROPEAN PATENT OFFICE	Dagmar Frank	

ANNEX TO THE INTERNATIONAL SEARCH REPORT
ON INTERNATIONAL PATENT APPLICATION NO.

US 9206142
SA 62640

This annex lists the patent family members relating to the patent documents cited in the above-mentioned international search report.
The members are as contained in the European Patent Office EDP file on
The European Patent Office is in no way liable for these particulars which are merely given for the purpose of information. 13/10/92

Patent document cited in search report	Publication date	Patent family member(s)	Publication date
FR-A-2626291	28-07-89	US-A- 5017336	21-05-91
		JP-A- 2004937	09-01-90
DE-A-3805124	31-08-89	US-A- 4938920	03-07-90
EP-A-0468093	29-01-92	US-A- 5080861	14-01-92
US-A-2872310		None	

EPO FORM P0119

For more details about this annex : see Official Journal of the European Patent Office, No. 12/82